

Photoluminescence and Crystalline Properties of CuO-Ta₂O₅ Composite Films Prepared Using Co-Sputtering

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Abstract

We prepared CuO-Ta₂O₅ composite films by our simple co-sputtering method using three CuO pellets and a Ta₂O₅ disc as a co-sputtering target, and subsequently annealed the films in ambient air at 900°C, 1000°C, and 1100°C for 20 min. We evaluated photoluminescence (PL) and X-ray diffraction properties of the annealed films, and discussed the relationship between sharp PL peaks ($\lambda \sim 450$ nm) observed from all the films and their crystallizabilities. We considered that the 450-nm peaks originated from Cu_{2.1}(Ta₄O₁₂) crystal phases in the films.

Keywords

Ta₂O₅, CuO, Co-Sputtering, Photoluminescence, X-Ray Diffraction

Subject Areas: Composite Material, Material Experiment

1. Introduction

Tantalum (V) oxide (Ta_2O_5) is a higher refractive index (n > 2) and lower phonon energy $(100 - 450 \text{ cm}^{-1})$ material than other popular oxides (e.g., SiO₂). It can be widely applicable to various passive or active optoelectronic elements such as anti-reflection coatings for silicon solar cells [1], photonic crystals prepared using the autocloning method [2] [3], and novel phosphors doped with rare-earths [4]. We have so far prepared various rare-earth (Er, Eu, Yb, Tm, Y, and Ce) doped Ta₂O₅ thin films using radio-frequency (RF) magnetron co-sputtering of rare-earth oxide (Er₂O₃, Eu₂O₃, Yb₂O₃, Tm₂O₃, Y₂O₃, and CeO₂) pellets and a Ta₂O₅ disc [5]-[18], and we have obtained various photoluminescence (PL) properties from the films.

Copper (Cu) is one of transition metals, and it is used as a functional dopant in light-emitting materials such as ZnS:Cu [19]-[21] and ZnO:Cu [22]. It is expected that novel Ta_2O_5 -based functional materials will be realized by doping with Cu instead of rare-earths into Ta_2O_5 . We have prepared Cu(II) oxide (CuO) and Ta_2O_5 co-

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sputtered (CuO-Ta₂O₅) composite films using a CuO pellet and a Ta₂O₅ disc as a co-sputtering target, and we have evaluated X-ray diffraction (XRD) and PL properties of the films after annealing at 600°C - 900°C [23]. In this short report, we will present the preparation of CuO-Ta₂O₅ composite films using more CuO pellets and a Ta₂O₅ disc as a co-sputtering target, and the evaluations of PL and XRD properties of the films annealed at higher temperatures of 900°C - 1100°C than those in our previous report [23]. Subsequently, we will discuss the relationship between the PL and XRD properties.

2. Experiments

A CuO-Ta₂O₅ film was deposited using our RF magnetron sputtering system (ULVAC, SH-350-SE). A schematic figure of the system was presented in our previous report [6]. A Ta₂O₅ disc (Furuuchi Chemical Corporation, 99.99% purity, diameter 100 mm) was installed as a sputtering target in the system. We placed three CuO pellets (Furuuchi Chemical Corporation, 99.9% purity, diameter 20 mm) on the erosion area of the Ta₂O₅ disc as presented in **Figure 1**. The flow rate of Ar gas introduced into the processing vacuum chamber was 15 sccm, and the pressure in the chamber during deposition was kept at ~5.4 × 10⁻⁴ Torr. The CuO pellets and the Ta₂O₅ disc were co-sputtered by supplying RF power to a cathode under the Ta₂O₅ disc. The RF power was set to 200 W. A fused-silica plate was used as a substrate, and it was not heated during deposition. We prepared four specimens from the as-deposited CuO-Ta₂O₅ sample by cutting it using a diamond-wire saw, and we subsequently annealed three of the four specimens in ambient air at 900°C, 1000°C, or 1100°C for 20 min using an electric furnace (Denken, KDF S-70).

The PL spectra of the three specimens were measured using a dual-grating monochromator (Roper Scientific, SpectraPro 2150i) and a CCD detector (Roper Scientific, Pixis:100B, electrically cooled to -80° C) under excitation using a He-Cd laser (Kimmon, IK3251R-F, wavelength (λ) 325 nm). The XRD patterns of the specimens were recorded using an X-ray diffractometer (RIGAKU, RINT2200VF+/PC system).

3. Results and Discussion

Figure 2 presents PL spectra of the three specimens annealed at 900°C, 1000°C, and 1100°C. Sharp PL peaks at $\lambda \sim 450$ nm were observed from all the specimens. The relative intensities of the PL peaks from the specimens annealed at 900°C, 1000°C, and 1100°C were 1, 1.86, and 2.14, respectively. **Figure 3** presents XRD patterns of the same specimens. Three significant diffraction peaks were observed from all the specimens. These peaks correspond to orthorhombic Cu_{2.1}(Ta₄O₁₂) ((0 0 2), (2 0 0), and (0 2 2)) phases (JCPDS No.01-076-7904). Therefore, the CuO-Ta₂O₅ films annealed at 900°C - 1100°C seems to have Cu_{2.1}(Ta₄O₁₂) crystal phases.

A similar PL peak at $\lambda \sim 450$ nm have already been observed only from an amorphous CuO-Ta₂O₅ film annealed at 600°C in our previous work [23]. The peak seemed to be attributed to the transition from the conduction band of Ta₂O₅ to the *t*₂ energy level of Cu²⁺ in the bang gap of Ta₂O₅ [19] [21] [23]. However, as mentioned above, the presented CuO-Ta₂O₅ films annealed at 900°C - 1100°C seemed to be not amorphous but partially Cu_{2.1}(Ta₄O₁₂) crystal phases. The origin of the sharp PL peaks presented in **Figure 2** may be different from that reported in [23]. In addition, we have reported that CuO-Ta₂O₅ films prepared using a CuO pellet become tetragonal CuTa₂O₆ phases after annealing at 700°C - 900°C, and no sharp PL peak was observed from the films



sputtering of three CuO pellets and a Ta_2O_5 disc.



Figure 2. PL spectra of CuO-Ta₂O₅ composite films annealed at 900°C, 1000°C, and 1100°C.



[23]. Therefore, the sharp 450-nm peaks observed from the CuO-Ta₂O₅ films annealed at 900°C - 1100°C seem to originate from the Cu_{2.1}(Ta₄O₁₂) phases in the films. We will continue to further investigate the origin of the 450-nm peaks by characterizing morphologies of the films using a scanning electron microscope.

4. Summary

We prepared CuO-Ta₂O₅ composite films by our simple co-sputtering method using three CuO pellets and a Ta₂O₅ disc as a co-sputtering target, and subsequently annealed the films in ambient air at 900°C, 1000°C, and 1100°C for 20 min. We evaluated PL and XRD properties of the annealed films, and discussed the relationship between sharp PL peaks ($\lambda \sim 450$ nm) observed from all the films and their crystallizabilities. From the results presented in this short report, we considered that the 450-nm peaks originated from Cu_{2.1}(Ta₄O₁₂) crystal phases in the films. Further investigations are going to be conducted in order to make the origin of the 450-nm peaks clearer.

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