

Kinetics of Phase Transformations and Thermal Stability of $Ge_xSe_{70}Sb_{30-x}$ (x = 5, 10, 15, 20) Chalcogenide Glasses

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Received May 6th, 2013; revised June 6th, 2013; accepted June 13th, 2013

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ABSTRACT

Kinetic studies of two-phase transformations *i.e.* glass transition and crystallization for $Ge_xSe_{70}Sb_{30-x}$ (x = 5, 10, 15, 20) glasses have been performed using differential scanning calorimetry (DSC) at different heating rates under non-iso-thermal condition. The glass transition and crystallization regions have been investigated in terms of their characteristic temperatures *i.e.* glass transition temperature T_g and crystallization temperature T_c and activation energy. The activation energies of glass transition and crystallization have been calculated by employing peak shift method of Kissinger and isoconversional methods. Activation energies have been found to be dependent on the composition. The thermal stability is determined from the temperature difference $\Delta T = T_c - T_g$ where T_c and T_g are the onset crystallization and glass transition temperature respectively. It has been found that $Ge_{15}Se_{70}Sb_{15}$ glass is thermally more stable as compared to the other members of the series.

Keywords: Glass Transition Region; Differential Scanning Calorimetry; Activation Energy; Thermal Stability

1. Introduction

Among amorphous semiconducting materials, chalcogenide glasses containing a large amount of chalcogen atoms sulphur, selinium and tellurium have drawn great attention because of their broad applications in technological field such as memory devices and fiber optics. Selinium is an attractive material in optical memory devices due to its uniqueness of reversible transformation [1]. But pure Se has certain disadvantages like short lifetime and low stability. To improve these parameters certain other elements are added [2]. The Ge-Se-Sb alloys have good glass formation ability over a wide compositional range. This makes them good candidates as arsenic free IR transparent materials [3,4]. The Ge-Se-Sb glasses are used for fabrication of optical components like IR lenses for their transparency in the 2 µm to 16 µm wavelength region [5-7]. The aim of the present paper is to analyze the dependence of characteristic temperatures T_g , T_c and the activation energy on heating rate as well as composition respectively. An effort has also been made to determine the thermal stability of Ge-Se-Sb glasses.

2. Experimental Details

High purity (99.999%) Germenium, Selinium and Antimony were weighed and the weighed materials are then introduced into clean quartz ampoules with appropriate atomic weight percentage. The contents of the ampoule were sealed in vaccum of 10^{-6} torr and then heated in furnace where temperature was raised at a rate of 3 - 4 K·min⁻¹ up to 925°C. The contents were kept around that temperature for 12 - 14 hours with continuous rotation to ensure the homogeneity of the sample. The melton sample was rapidly quenched in ice-cooled water to get glassy state. The ingot of so produced glassy samples were taken out of the ampoule by breaking the ampoule and then grinded gently in mortar and pestle to obtain powder form.

The glasses are amorphous material therefore amorphous nature of these samples was characterized by X-ray diffraction pattern using Bragg-Brentano geometry on a Panalytical X'pert Diffractometer with a Cu K α radiation source (1.5406 Å).

The calorimetric measurements were carried out using differential scanning calorimetery (Rigaku DSC-8230)

with an accuracy of ± 0.1 K. 10 mg powdered samples were crimped into aluminium pans and scanned at different (10, 15, 20, 25 and 30 K·min⁻¹) heating rates.

3. Result and Discussion

Figure 1 shows the DSC thermograms of $Ge_xSe_{70}Sb_{30-x}$ (x = 5, 10, 15, 20) chalcogenide glasses at a heating rate of 10 K·min⁻¹. It has been observed that each thermogram except for $Ge_{20}Se_{70}Sb_{10}$ represents two well-defined peaks. The first one is endothermic peak correspon ding to the glass transition region and other is exothermic peak corresponding to crystallization region. **Figure 2** shows the DSC thermograms of $Ge_5Se_{70}Sb_{25}$ with different heating rates (10, 15, 20, 25 and 30 K·min⁻¹). It has been observed from **Figure 2** that both glass transition and crystallization temperatures shift towards the higher temperature side with the increase of heating rate.

3.1. Kinetics of Glass Transition Region

The study of the glass transition region for Ge_xSe₇₀Sb_{30-x}



Figure 1. DSC thermograms of $Ge_xSe_{70}Sb_{30-x}$ (x = 5, 10, 15, 20) chalcogenide glasses at a heating rate of 10 K·min⁻¹.



Figure 2. DSC thermograms of $Ge_5Se_{70}Sb_{25}$ with different heating rates (10, 15, 20, 25 and 30 K·min⁻¹).

(x = 5, 10, 15, 20) alloys has been made in terms of variation of glass transition temperature and activation energy with different heating rates and different compositions. The values of onset and peak glass transition and crystallization temperatures for all the compositions of the glassy materials at a heating rate of 10 K·min⁻¹are given in **Table 1** respectively.

The glass transition temperature represents the strength of the glassy structure of the alloy. The variation of glass transition temperature with heating rate (β) has been explained on the basis of empirical relation proposed by Lasocka [8] as given below.

$$T_{\sigma} = A + BLn\beta \tag{1}$$

where *A* and *B* are constants for a given sample. The value of *A* indicates the glass transition temperature for the heating rate 1 K·min⁻¹ and B indicates time which is taken by the system to reduce the glass transition temperature when heating rate decreased from 10 K·min⁻¹ to1 K·min⁻¹. Plots of T_g versus Ln β are shown in **Figure 3** for Ge_xSe₇₀Sb_{30-x} (x = 5, 10, 15, 20) glassy samples. The slopes and intercepts of the plots (straight lines) give the values of B and A respectively. The values of A and B for different compositions of glassy alloys are given in **Table 2**. It is obvious from **Figure 3** that T_g increases for all the values of x with increasing heating rate.



Figure 3. Plots of T_g versus $Ln\beta$.

Table 1. Values of characteristic temperatures for $Ge_xSe_{70}Sb_{30-x}$ glassy alloys at a heating rate of 10 K·min⁻¹.

Sb at wt%	Characteristic temperatures for $Ge_xSe_{70}Sb_{30-x}$ glassy alloys at a heating rate of 10 K·min ⁻¹			
	$T_{g}\left(\mathbf{K}\right)$	$T_{gp}\left(\mathbf{K}\right)$	$T_{c}\left(\mathbf{K}\right)$	$T_{p}\left(\mathbf{K}\right)$
10	479	506	-	-
15	452	474	516	629
20	475	494	533	575
25	444	463	483	511

The glass transition region consists of various metastable states separated by energy barriers. The atoms in these metastable states tend to attain more stable state than others by overcoming this energy barrier. This energy barrier is known as activation energy of glass transition region. The atoms in a glass having minimum activation energy have higher probability to jump to the metastable state of lower energy which is the most stable state [9].

The activation energy of glass transition region has been evaluated using Kissinger formulation [10]. The Kissinger formulation is basically used for the calculation of the activation energy of crystallization region. It has also been frequently used for the evaluation of the activation energy of glass transition region [11-13]. The Kissinger equation is

$$Ln(\beta/T_{gp}^2) = \text{constant} - E_t/RT_{gp}$$
 (2)

where T_{gp} is the temperature of endothermic peak and R is universal gas constant. Plots of $Ln(\beta/T_{gp}^2)$ versus $1000/T_{gp}$ have been shown in **Figure 4** for Ge_xSe₇₀Sb_{30-x} (x = 5, 10, 15, 20) glassy sample. The activation energy of glass transition has been determined from the slope of curve. The activation energy of glass transition for different composition of glasses is given in **Table 3**. It is observed from **Table 3** that the activation energy is minimum for 15 at wt% of Sb.





 Table 2. Values of A and B for different compositions of Ge-Se-Sb system.

Sb at wt%	<i>A</i> (K)	B (min)
10	458.3	8.70
15	429.7	9.33
20	450.1	10.35
25	415.8	12.30

Table 3. Values of the activation energy of glass transition	on E_t
for $Ge_x Se_{70}Sb_{30-x}$ (x = 5, 10, 15, 20) glassy samples.	

Activation energy of glass transition E_t (KJ·mole ⁻¹) Kissinger method	
235.28	
197.28	
409.16	
258.03	

3.2. Kinetics of Crystallization

Kinetics of crystallization process under non-isothermal condition has been analyzed in terms of composition dependence of onset crystallization temperature (T_c) , peak crystallization temperature and the activation energy of crystallization region. The activation energies to be considered in an amorphous crystalline transformation process are the activation energy for nucleation (E_n) , activation energy for crystal growth (E_{α}) and that whole process of crystallization, called the activation energy of crystallization denoted by E_c . The activation energy of crystallization under non-isothermal conditions for $Ge_x Se_{70}Sb_{30-x}$ (x = 5, 10, 15) has been evaluated from two different methods. It would have not been possible to evaluate the activation energy of crystallization for Ge₂₀Se₇₀Sb₁₀ as the onset and peak crystallization temperature have not been obtained in these glasses at any heating rates.

First one is Peak method; the activation energy of crystallization can be obtained from the dependence of T_p on different heating rates, using the equation of Kissinger model [14]

$$Ln(\beta/T_p^2) = \text{constant} - E_c/RT_p$$
 (3)

where T_p is the peak crystallization temperature. Plots of $Ln(\beta/T_p^2)$ versus $1000/T_p$ are shown in **Figure 5** for $Ge_xSe_{70}Sb_{30-x}$ (x = 5, 10, 15) glassy samples. As mentioned above the slope of the plots (straight lines) give the activation energy of crystallization.

The second method for the determination of the activation energy of crystallization is isoconversional method. Isoconversional method has two types. One set of methods relies on approximating the temperature integral such as Kissinger-Akahira-Sunose (KAS) [13,14] and Flynn-Wall-Ozawa method [15,16]. The second set of methods uses a determination of the reaction rate at an equivalent stage of the reaction at different heating rates such as Frideman-Ozawa method [17]. In the present paper, the activation energy of crystallization has been calculated using Kissinger-Akahira-Sunose (KAS) and Flynn-Wall-Ozawa isoconversional methods.

3.3. Kissinger-Akahira-Sunose (KAS) Method

$$Ln(\beta/T_{\alpha i}^{2}) = \text{constant} - E_{c}/RT_{\alpha i}$$
(4)

where T_{ai} temperature at which a fixed fraction α of the total amount is transformed. Plots of $Ln(\beta/T_{\alpha i}^2)$ versus $1000/T_{ai}$ are shown in **Figure 6** for Ge_xSe₇₀Sb_{30-x} (x = 5, 10, 15) glassy samples. Again the slope of the straight line gives the activation energy of crystallization using this method.

3.4. Flynn-Wall-Ozawa (FWO) Method

$$Ln(\beta) = \text{constant} - 1.0518 E_c / RT_{\alpha i}$$
(5)

where T_{ai} temperature at which a fixed fraction α of the total amount is transformed. **Figure 7** shows the Plots of $Ln(\beta)$ versus $1000/T_{ai}$ for Ge_xSe₇₀Sb_{30-x} (x = 5, 10, 15) glassy samples. The activation energy of crystallization has been determined from the slope of the straight line.

The activation energy of crystallization E_c is an indication of the speed of the rate of crystallization. **Table 4** shows the activation energy of crystallization for differrent composition of glasses and it has been noticed that the activation energy is highest for 15 at wt% of Sb.



Figure 5. Plots of $Ln(\beta/T_p^2)$ versus $1000/T_p$.

Table 4. Values of the activation energy of crystallization E_c for different compositions.

Sb at wt% -	Activation energy of crystallization E_c (KJ·mole ⁻¹)			
	Kissinger method	KAS method	FWO method	
10	-	-	-	
15	175.08	162.34	164.53	
20	158.61	155.15	156.88	
25	152.59	151.86	152.62	



Figure 6. Plots of $Ln(\beta/T_{ai}^2)$ versus $1000/T_{ai}$.



Figure 7. Plots of $Ln(\beta)$ versus $1000/T_{\alpha i}$.

3.5. Thermal Stability

From application point of view the glass should be thermally stable. The thermal stability of the glasses has been decided on the basis of kinetic resistance to crystallization. The kinetic resistance is the difference of the onset crystallization (T_c) temperature and onset glass transition temperature (T_g). The larger the difference then higher is the resistance [18]. Therefore, in Ge_xSe₇₀Sb_{30-x} (x = 5, 10, 15, 20) glasses a composition, which has higher value of kinetics resistance, is most stable. It has been observed that glass Ge₁₅Se₇₀Sb₁₅ has the highest thermal stability. **Figure 8** shows the variation of $T_c - T_g$ with percentage composition of Sb in the Ge-Se-Sb glasses. It is also noted from **Table 1** that $T_c - T_g$ is highest for 15 at wt% of Sb in the glass and therefore Ge₁₅Se₇₀Sb₁₅ glass is thermally more stable.

4. Conclusions

The studies of kinetics of crystallization and thermal sta-



Figure 8. Variation of $T_c - T_g$ with atomic weight percentage of Sb.

bility of $Ge_x Se_{70}Sb_{30-x}$ (x = 5, 10, 15, 20) chalcogenide glasses have led to following conclusions:

- Glass transition and crystallization process have been shown in each DSC thermogram except Ge₂₀Se₇₀Sb₁₀ alloys.
- The glass transition temperature is found to vary for different composition indicating a structural change due to addition of Sb.
- A glassy sample having high kinetic resistance to crystallization is more stable as compared to other glasses. Accordingly Ge₁₅Se₇₀Sb₁₅ is more stable glass than the other members of the series.

5. Acknowledgements

One of the authors (Anusaiya) is grateful to the DSA programme of UGC in the department for providing financial support in the form of fellowship during this work.

REFERENCES

- K. Tanaka, "Structural Phase Transition in Chalcogenide Glasses," *Physical Review B*, Vol. 39, No. 2, 1989, pp. 1270-1279. <u>http://dx.doi.org/10.1103/PhysRevB.39.1270</u>
- [2] S. Kumar, M. Hussain and M. Zulfequar, "Effect of Silver on the Dielectric Properties," *Physica B*, Vol. 371, No. 2, 2006, pp. 193-198. http://dx.doi.org/10.1016/j.physb.2005.09.021
- [3] M. Esquerre, J. C. Carballes, J. P. Audiere and C. Mazieres, "Crystallization of Amorphos (Bulk and Thin Films) Ge_xSe_{1-x} (0 < x < 0.20) Alloys," *Journal of Materials Science*, Vol. 13, No. 6, 1978, pp. 1217-1223. <u>http://dx.doi.org/10.1007/BF00544727</u>
- [4] A. B. Ghokle and R. B. Abbaschaian, "The Ge-Se (Germenium-Selinum) System," *Bulletin of Alloy Phase Diagrams*, Vol. 11, No. 3, 1990, pp. 257-263. http://dx.doi.org/10.1007/BF03029295

- [5] A. R. Hillton and D. J. Hayes, "The Interdependence of Physical Parameters for Infrared Transmiting Glasses," *Journal of Non-Crystalline Solids*, Vol. 17, No. 3, 1975, pp. 339-348.
- [6] M. Frumer, H. Ticha, J. Klikoka and P. Tomiska, "Optical Absorption in Vitreous GeSb₂Se₄," *Journal of Non-Crystalline Solids*, Vol. 13, No. 1, 1973, pp. 173-178. <u>http://dx.doi.org/10.1016/0022-3093(73)90046-X</u>
- [7] J. A. Savage, P. J. Webber and A. M. Pitt, "An Assessment of Ge-Sb-Se Glasses as 8 to 12 μm Infrared Optical Materials," *Journal of Materials Science*, Vol. 13, No. 4, 1987, pp. 859-864. http://dx.doi.org/10.1007/BF00570524
- [8] M. Lasocka, "The Effect of Scanning Rate on Glass Transition Temperature of Splat-Cooled Te₈₅Ge₁₅," *Materials Science and Engineering*, Vol. 23, No. 2-3, 1976, pp. 173-176. <u>http://dx.doi.org/10.1016/0025-5416(76)90189-0</u>
- [9] M. M. A. Imran, D. Bhandri and N. S. Saxena, "Enthalpy Recovery during Structural Relaxations of Se₉₆In₄ Chalcogenide Glasses," *Physica B*, Vol. 293, No. 3-4, 2001, pp. 394-401. <u>http://dx.doi.org/10.1016/S0921-4526(00)00543-3</u>
- [10] H. E. Kissinger, "Reaction Kinetics in the Differential Thermal Analysis," *Analytical Chemistry*, Vol. 29, No. 11, 1957, pp. 1702-1706. http://dx.doi.org/10.1021/ac60131a045
- [11] H. S. Chen, "A Method for Evaluating Viscosities of Metallic Glasses from the Rates of Thermal Transformations," *Journal of Non-Crystalline Solids*, Vol. 27, No. 2, 1978, pp. 257-263. http://dx.doi.org/10.1016/0022-3093(78)90128-X
- [12] J. E. Shelby, "Thermal Expansion of Amorphous Metals," *Journal of Non-Crystalline Solids*, Vol. 34, No. 1, 1979, pp. 111-119. <u>http://dx.doi.org/10.1016/0022-3093(79)90010-3</u>
- [13] H. E. Kissinger, "Variation of Peak Temperature with Heating Rate in Differential Thermal Analysis," *Journal* of Research of the National Bureau of Standards, Vol. 57, No. 4, 1956, pp. 217-221. http://dx.doi.org/10.6028/jres.057.026
- [14] T. Akahira and T. Sunose, "Joint Convention of Four Electrical Institutes," *Science Technology*, Vol. 16, 1971, pp. 22-31.
- [15] T. Ozawa, "A New Method of Analyzing Thermogravimetric Data," *Bulletin of The Chemical Society of Japan*, Vol. 38, No. 11, 1965, pp. 1881-1886. <u>http://dx.doi.org/10.1246/bcsj.38.1881</u>
- [16] J. Flynn and A. Wall, "A Quick, Direct Method for the Determination of Activation Energy from Thermogravimetric Data," *Journal of Polymer Science Part B*, Vol. 4, No. 5, 1966, pp. 323-328. <u>http://dx.doi.org/10.1002/pol.1966.110040504</u>
- [17] H. L. Friedman, "Kinetics of Thermal Degradation of Charforming Plastics from Hermogravimetry and Application to a Phenolic Plastic," *Journal of Polymer Science Part C*, Vol. 6, No. 1, 1964, pp. 183-185.
- [18] S. Mahadevan, A. Giridhar and A. K. Singh, "Calorimetric Measurements on As-Sb-Se Glasses," *Journal of Non-Crystalline Solids*, Vol. 88, No. 1, 1986, pp. 11-34. http://dx.doi.org/10.1016/S0022-3093(86)80084-9