Mechanochemical Synthesis of Nano Calcium Silicate Particles at Room Temperature

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ABSTRACT

Nano-sized calcium silicate powders were synthesized at room temperature by the new mechanochemical method using a high energy planetary ball mill. The formation of calcium silicate from its raw materials (calcium carbonate and dehydrated silica gel) was monitored by the XRD analysis with progression of ball milling. It is observed that the synthetic process comes to an end through the following three sequential stages: comminution of raw materials, recombination of comminuted raw materials to final product, and comminution of final product to smaller sizes. The nanostructure of the synthesized powder was realized by the FESEM photomicrograph, TEM image and XRD analyses. These analytical observations have revealed that the nano-sized polycrystalline calcium silicate particles are formed after about 6 h of ball milling and they are spheroidal in shape. The average particle size of the as-generated calcium silicate nanocrystalline powders is found to be around 21 nm which decreases with increasing ball milling but increases with annealing at elevated temperature.

Keywords: Mechanochemical Processing, Calcium Silicate Nanoparticle, XRD, FESEM, TEM

1. Introduction

Calcium silicate has received significant attention because of their potential applications as host of phosphors and ceramic insulators [1-3]. Nanosized calcium silicate material has been studied over the past years because of their significant properties which are different from their bigger sizes. Conventionally, calcium silicate is prepared by the solid state reaction of CaO or CaCO3 and quartz (SiO2) at 1150°C - 1200°C for several hours [1-2]. It is also prepared by the chemical methods such as combustion, sol-gel, co-precipitation, etc. routes followed by heat treatment at various temperatures [3-5]. All these methods are limited by some of their inherent disadvantages, for example, requirement of high temperature, long processing time, low yield or hazardous to health and environment.

Recently, synthesis of nanomaterials by mechanochemical route opens up a new page in the nanoscience and nanotechnology. It is a quite simple, energy saving, low-cost, industrially scalable and viable non-conventional technique for the preparation of homogeneous and nano-sized (1 - 100 nm) multicomponent materials [6-9]. It is an environmentally friendly solvent-free and waste-free synthetic route as well unlike chemical methods. The main advantage of the mechanochemical method is that the solid-state reactions are carried out at room temperature instead of high temperature. Practically, it is a high energy milling process that involves repeated mixing, deformation, comminuting, welding and re-welding of the reactant powder particles in a closed vial of a planetary ball mill [10-13]. Here, the solid-state reactions are progressed by the kinetic energy transferred from the milling balls to the milled powders. Its versatility gives the impressions that the mechanochemical method is a promising novel technique for the synthesis of various advanced nanopowder materials. To the best of our knowledge, the synthesis of calcium silicate nanoparticles by the mechanochemical method has not been explored previously.

In this letter, we demonstrate the mechanochemical synthesis of calcium silicate nanoparticles from the calcium carbonate and silica gel at room temperature without applying additional heat treatment. Its synthesis is recognized by the FESEM photomicrograph, TEM image and XRD analyses.
2. Experimental Procedure

The starting raw materials were high purity calcium carbonate, CaCO₃ (99%, Fluka, Switzerland) and dehydrated silica gel, SiO₂ (99%, Fluka, Switzerland). A stoichiometric mixture of these raw materials equivalent to 5 g calcium silicate was ball milled in a 500 ml zirconia jar with 100 numbers of 10 mm diameter zirconia milling balls at 300 rpm in a high energy planetary ball mill (PM 100, Retsch). Separate samples were prepared by varying the milling duration such as 1, 3, 6, 10, 15, 20 and 26 h. One sample of 26 h duration was annealed at 750°C for 10 h in air to confirm the completion of synthesis by the mechanochemical method and also to observe the effect of heating on the mechanochemically synthesized calcium silicate at elevated temperature.

The XRD patterns of the starting raw materials mixture and as-milled product samples were recorded at 25°C with X’pert Pro MPD diffractometer (PANalytical) operating at 45 kV and 35 mA using Ni-filtered CuKα (λ = 1.5406 Å) radiation and the X’celerator with step size 0.05° (2θ) and step time 0.5 sec from 10° to 80°. FESEM photomicrographs were recorded with a Gemini Zeiss Supra™ 35VP Model (Carl Zeiss) instrument using an accelerating voltage of 4.9 kV. TEM images were taken using a FEI instrument (TECHNAI G2) operating at the accelerating voltage of 300 kV.

3. Results and Discussion

Calcium silicate (CaSiO₃) is synthesized from the raw materials, calcium carbonate (CaCO₃) and dehydrated silica (SiO₂) gel, by the mechanochemical method using a high energy planetary ball mill. Its solid state synthesis reaction which occurred in this method at room temperature could be represented by

\[
\text{CaCO}_3 + \text{SiO}_2 \rightarrow \text{CaSiO}_3 + \text{CO}_2
\]  

(1)

The synthesis of nano calcium silicate particles has found to be progressed in this method through the following three sequential stages. It is also more clearly depicted in Figure 1.

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Mixture of raw materials → Comminution of raw materials (Stage-I) → Recommination of comminuted raw materials to final product (Stage-II) → Comminution of final product to smaller sizes (Stage-III)

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The effects of milling time on the generation and subsequent comminution of calcium silicate are monitored by the XRD patterns by comparing with its standard XRD pattern (JCPDS File Card No. 27-0088). These are shown in Figure 2. The particle sizes were calculated using the following Scherrer’s formula [14]:

\[
D = \frac{0.9\lambda}{B \cos 2\theta} \quad \text{(peak)}
\]  

(2)

where \(D\) is the diameter of the calcium silicate particle, \(\lambda\) is wavelength of CuKα radiation (\(\lambda = 1.5406\ \text{Å}\)), \(B\) is full width at half maximum (FWHM) of the intensity peak and \(2\theta\) is diffraction angle at maxima of high intensity peak in the patterns.

Figure 2 shows that during early stage of milling (up to 3 h) there is no evidence of calcium silicate formation but only the comminution of starting raw materials, e.g., calcium carbonate, down to 11 nm. Then after the particle size instantly increases (21 nm) up to 6 h of milling with generation of calcium silicate. Between 3 and 6 h of milling the qualitative change of XRD patterns due to calcium silicate formation is clearly visible. This observation shows that during this period of milling the particles are severely deformed, amorphized and re-combined due to high energy impacts of the milling balls. At 6 h of milling, it clearly shows the calcium silicate peaks which confirm the formation of calcium silicate crystalline phase by the mechanochemical method. Subsequently, the particle size of the as-generated calcium silicate decreases down to 13 nm and the peak intensities increases with increase in milling time, that is, 6 h onward. Thus, the synthesis of more calcium silicate gradually increases along with reduction of its sizes with increasing milling time.

Further qualitative changes are observed after 10 h of milling in the XRD patterns. It shows more prominent development of <320>, <432> and <921> hkl planes of calcium silicate (JCPDS File Card No. 27-0088). The information obtained from XRD patterns analyses agree well with the field emission scanning electron microscopy (FESEM) photomicrograph as shown in Figure 3(a). FESEM observations revealed the material morphology with milling time by mechanochemical treatment. The as-prepared sample after 10 h milling indicates that the synthesized calcium silicate particles are joined with each other due to recombination reactions thereby increases the sizes as observed in the XRD. This is already shown in the Figure 1. The transmission electron microscopy (TEM) image of the neck of two particles and its SAED pattern are shown in Figure 4(a). The formation of calcium silicate was started around 6 h of milling. Further milling, the calcium silicate gradually developed broad XRD peaks without any characteristic changes of the
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Figure 1. The particle size as a function of milling time.

Figure 2. X-ray diffraction patterns of samples milled for different duration and the 26 h milled sample annealed at 750°C (Top). The standard pattern (JCPDS File Card No. 27-0088) of calcium silicate (CS) is also shown for comparison (Bottom).

patterns which point towards the gradual reduction of its particle size with increasing milling time, as realized by evaluating using the Equation (2).

The main changes occur after 26 h of milling is that its particle sizes reduced to about 13 nm as calculated from XRD. The morphological changes of 26 h milled calcium silicate particles are also clearly visible in the FESEM photomicrograph as shown in Figure 3(b). The synthesized calcium silicates separated from each other and get their defined morphological shapes which are spheroidal in nature. The transmission electron microscope (TEM) image and the SAED pattern, as shown in Figure 4(b), illustrate the polycrystallinity of the mechanochemically synthesized calcium silicate. The TEM image also shows the rough and irregular characteristics of the surfaces of calcium silicate particles. It is very difficult to conclude

the same with only the FESEM photomicrograph as shown in the Figure 3(b). However, the TEM image gives the unambiguous view.

We have annealed the synthesized calcium silicate after 26 h of milling at 750°C for 10 h in an electrical furnace to confirm the completion of synthesis of calcium silicate by the mechanochemical method at 6 h or onward. Its XRD pattern is shown at the top of the Figure 2. It is seen that the width of peaks of the XRD pattern of the annealed powder reduces with increase in their intensities but without any additional peak. Thus, it indicates only the growing of the calcium silicate particle size due to fusion or sintering of smaller particles on annealing at 750°C. The heat treated powders do not exhibit any new XRD peak which clearly depicts that the solid state reaction has completed in the mechanochemical process and no further new compound is formed even though after application of high temperature.

4. Conclusions

Synthesis of calcium silicate nanoparticles at room temperature by the new mechanochemical method has been demonstrated. The formation of calcium silicate has been recognized by the FESEM photomicrograph, TEM image and XRD analyses. It is observed that the synthetic process comes to an end through the following

Figure 3. FESEM photomicrographs of samples obtained after (a) 10 and (b) 26 h of milling.
Three sequential stages: comminution of raw materials, recombination of comminuted raw materials to final product, and comminution of final product to smaller sizes. The nano-sized polycrystalline calcium silicate particles are found to generate after about 6 h of ball milling and they are spheroidal in shape. The average particle size of the as-generated particles is found to be around 21 nm which decreases down to 13 nm with increasing ball milling but increases up to 30 nm after annealing at 750°C. The annealed sample does not show any new XRD peaks, therefore, it authenticates the completion of the solid state synthesis reactions of calcium silicate at room temperature by the mechanochemical method using high energy planetary ball milling. We believe that this work would open a new vista in the area of nanoparticle research.

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REFERENCES


